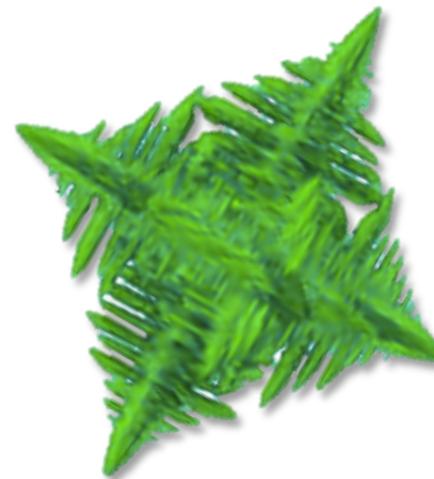


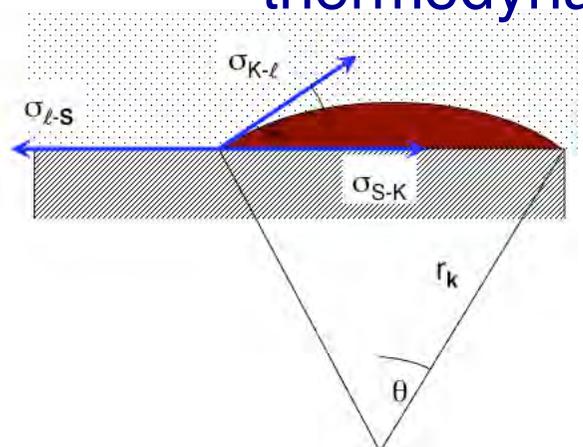
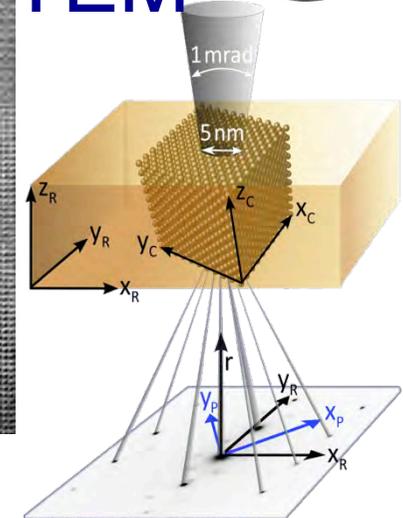
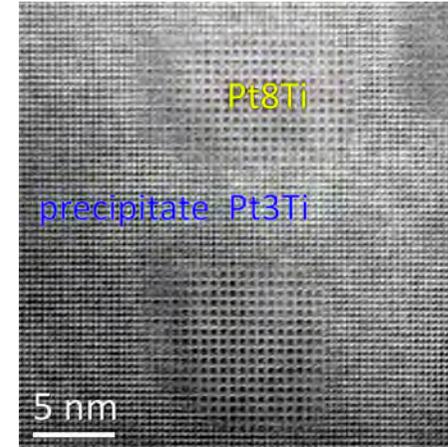
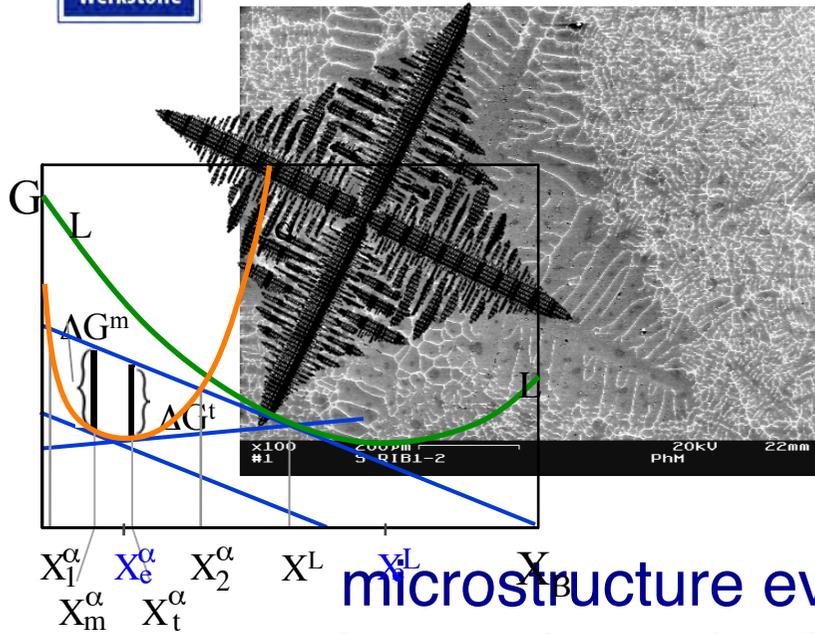
Modeling of dendritic growth – are there alternatives to the phase field method?

Markus Rettenmayr, Klemens Reuther
Friedrich Schiller University Jena
Otto Schott Institute of Materials Research
Chair of Metallic Materials

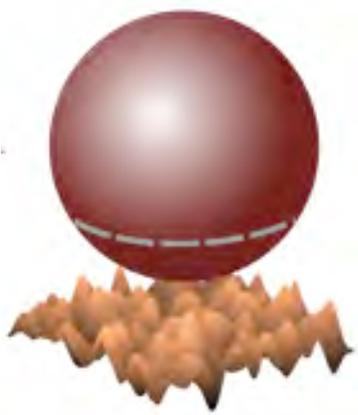




TEM



interfaces

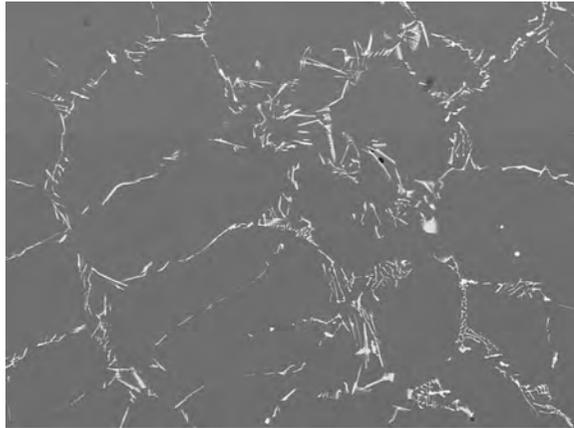


alloy development

acknowledgements

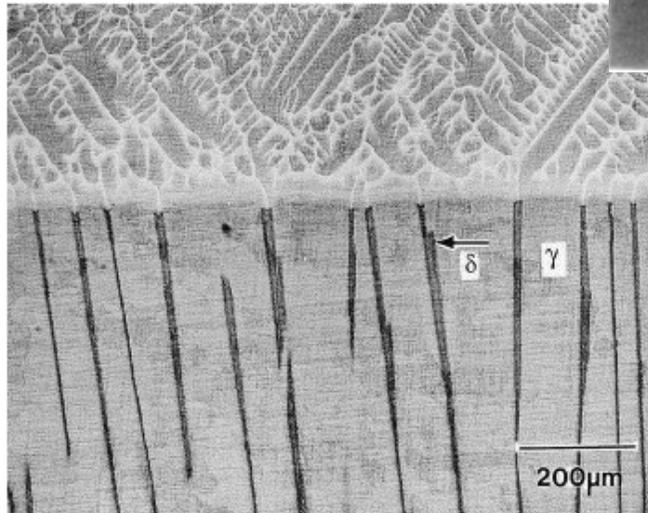
H. Lauterbach
Dr. R. Sumathi
Dr. Roschen Sasikumar
Prof. Bozidar Sarler
Prof. Michel Rappaz
Prof. Hans Eckart Exner †
Prof. Martin Glicksman



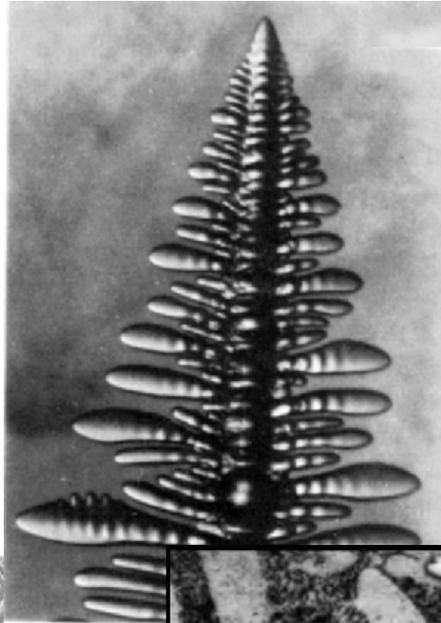


Al-Fe-Si

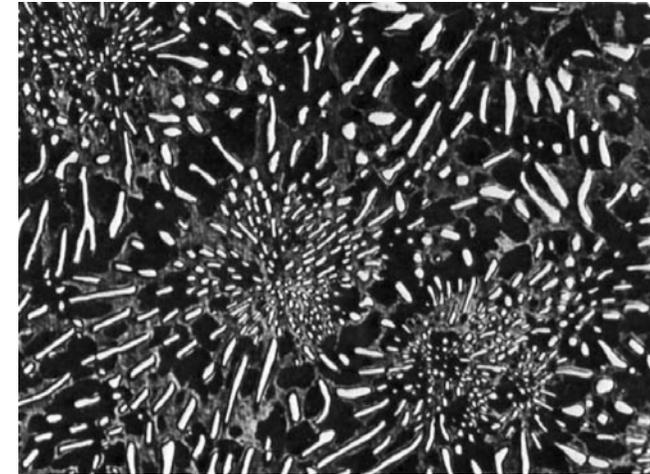
Fe-Ni



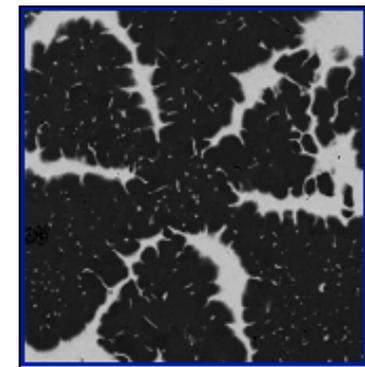
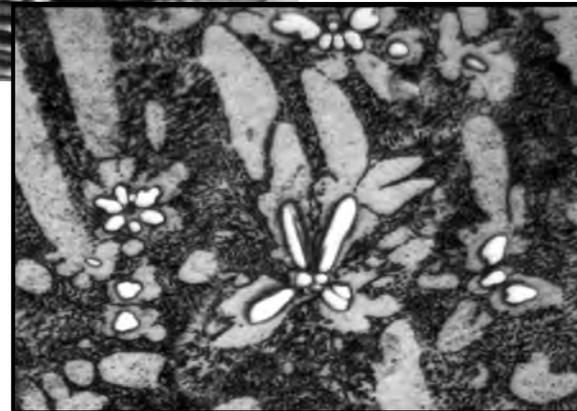
SCN



Cd-Cu-Ag



Pb-Sn-Ag-Cu



Ag-Bi



10^{14} dendrites in castings solidify worldwide per second

size: micrometers to meters

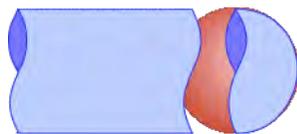
growth in crystallographic direction

→ cubic lattice: 4-fold symmetry

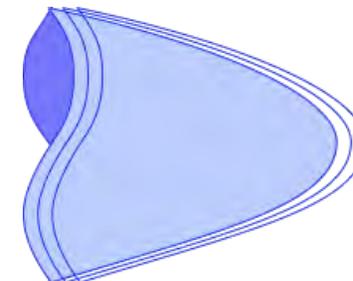
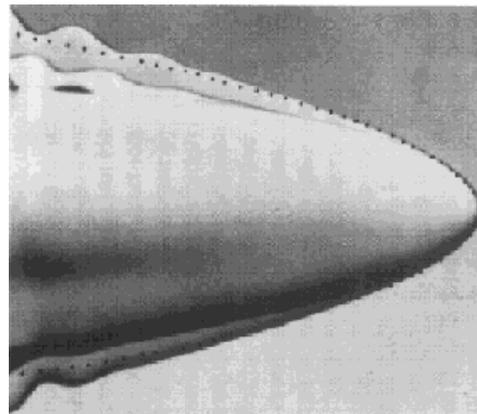


experimental observation: dendrite tip = paraboloid of revolution

first analytical model

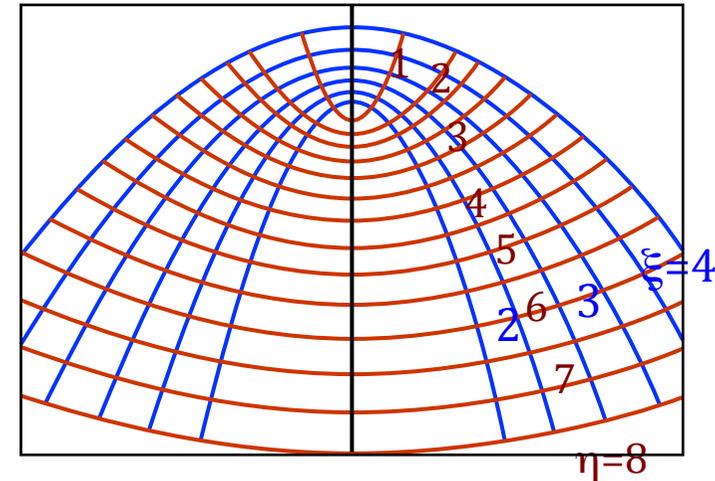
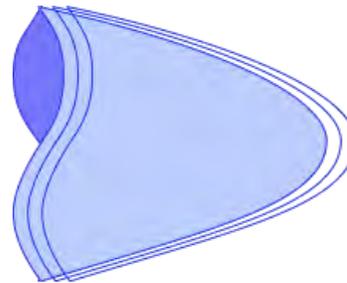
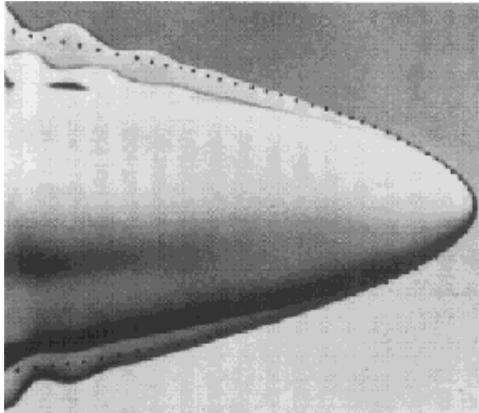


'hemispherical needle'





diffusion at a dendrite tip



analytical solution 1940ies:

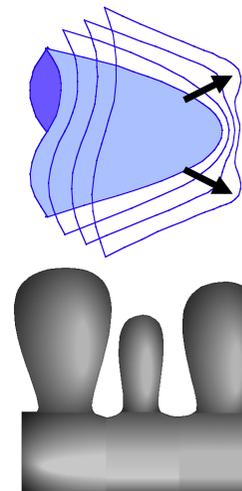
”Ivantsov transport solution“

(parabolic coordinate system)

influence of interfacial energy not included

→ tip shape not realistic

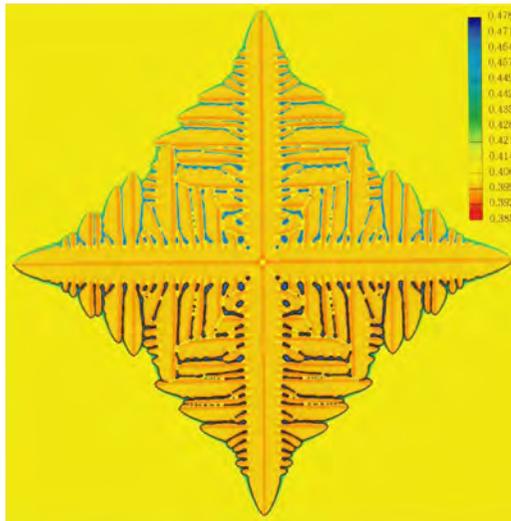
→ secondary arms not at all included



solves diffusion equation in the vicinity of a moving boundary

describes complex morphologies

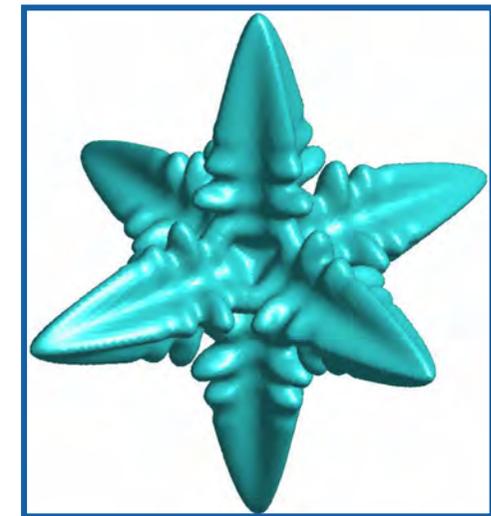
includes interface energy and its anisotropy



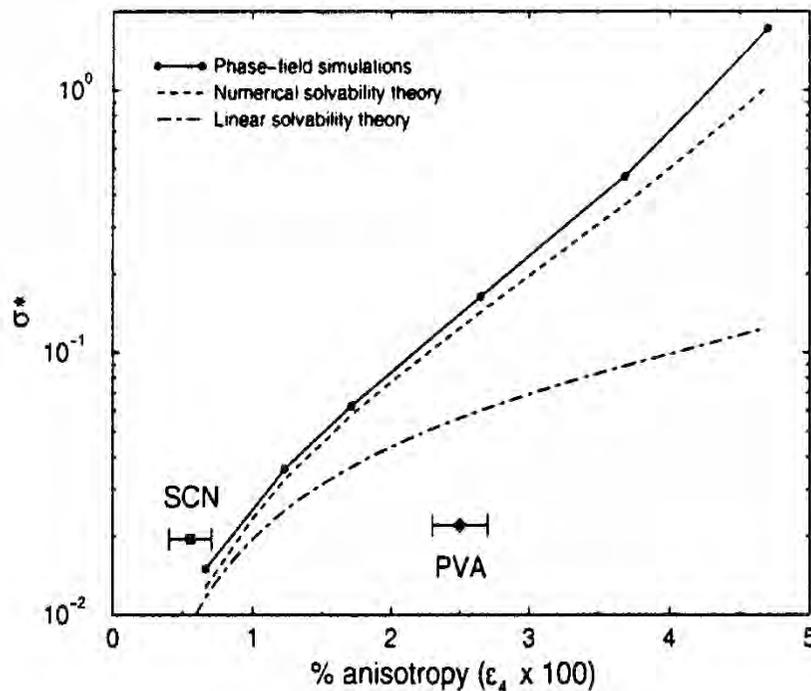
J. Warren, W.Boettinger



René Magritte



A. Karma, M. Rappaz



physical interface thickness only in 1D
(otherwise simulation too slow)

→ thicker interface, "anti-trapping current"

weak grid anisotropy remains

→ empirical corrections, choice of grid

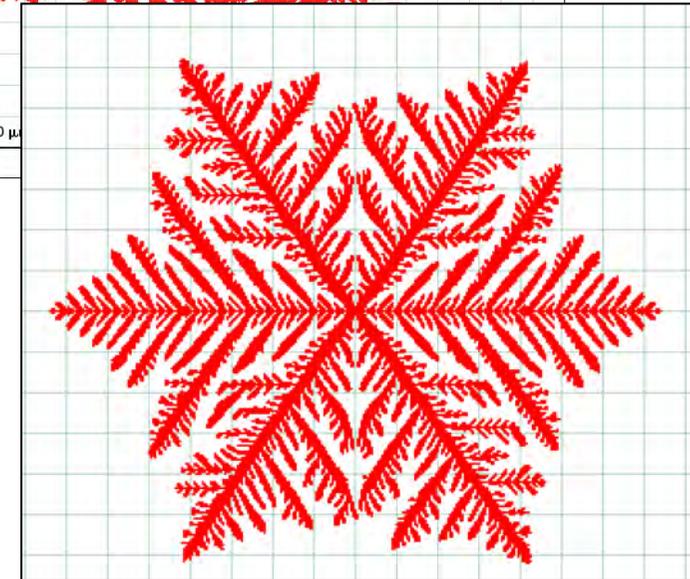
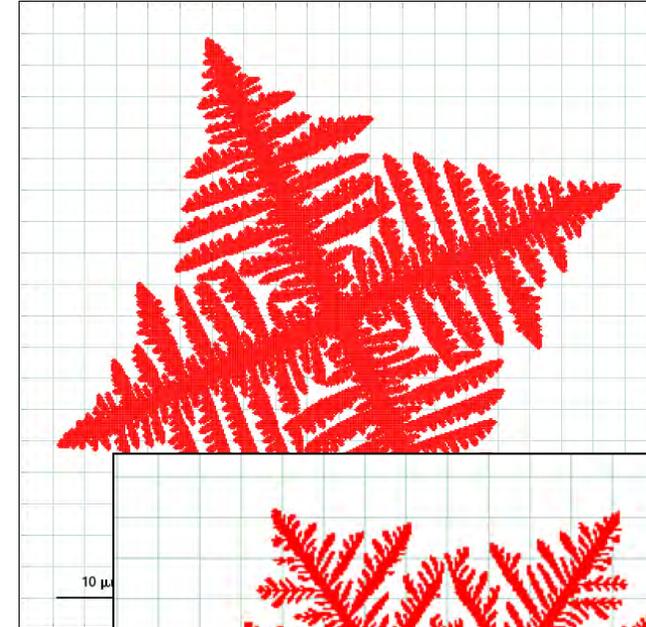
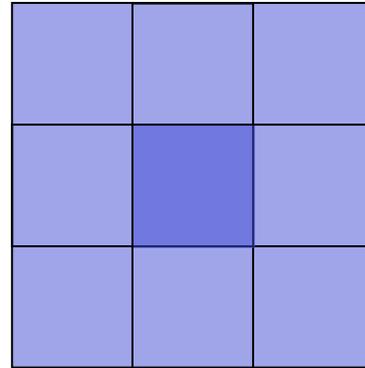
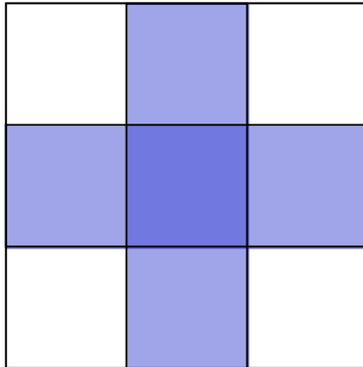
high computational cost

→ avoid slow (technical) processes

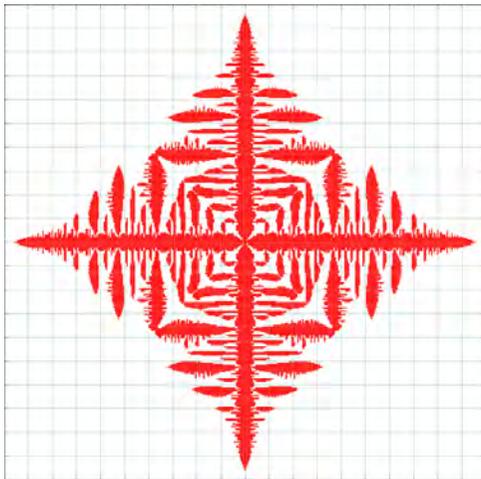
interface energy anisotropy considered,

but not in agreement with experiments

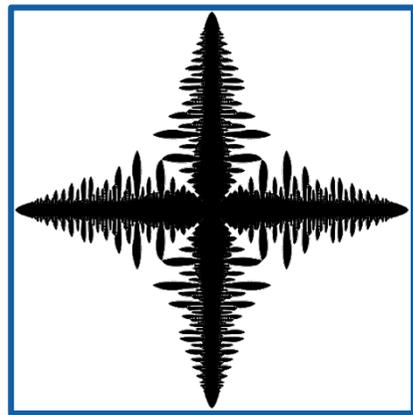
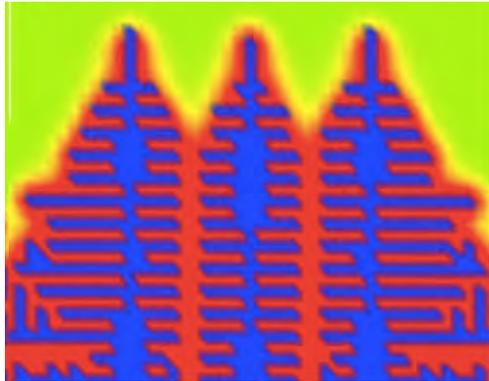
→ attack experimentalists



von Neumann or *Moore* neighborhoods
very very strong grid anisotropy

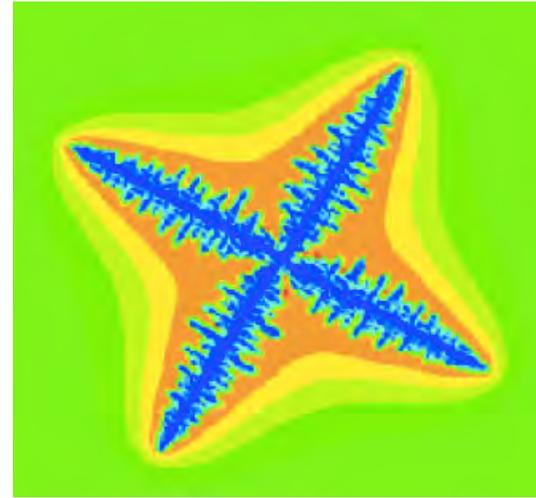
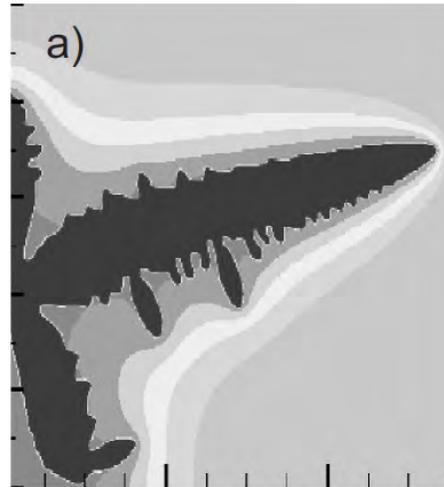


hybrid neighborhoods ? → work of A. Lorbiecka, B. Sarler



- level set method for interface position
- sophisticated curvature treatment

CA dendrite, K. Reuther, M. Rettenmayr, Comp. Mater. Sci. 2012

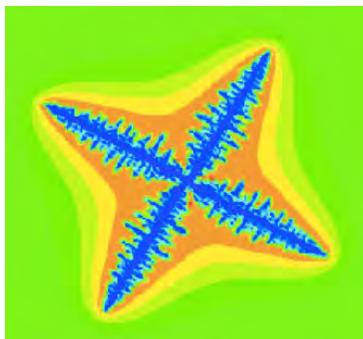
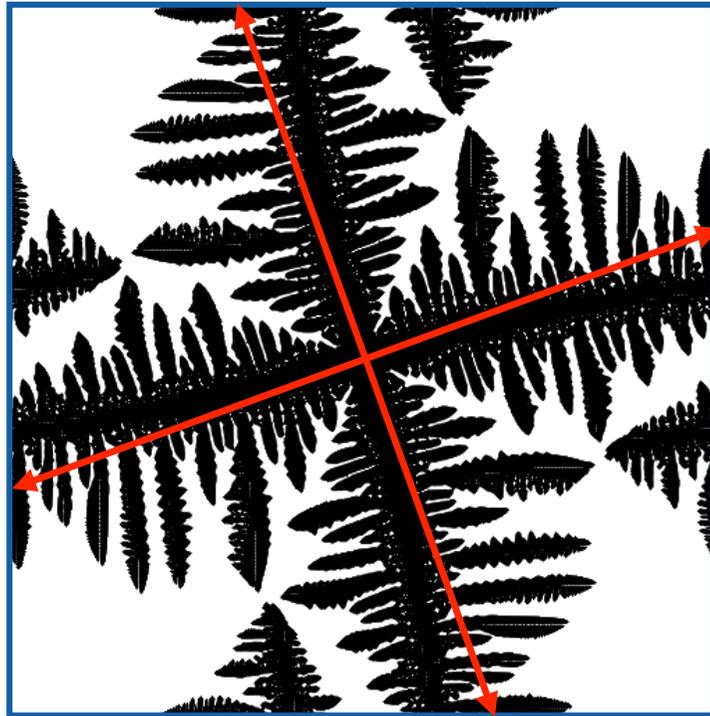


grid with mesh size of $\approx 1\mu\text{m}$

growth in **any** direction with respect to the grid*

arm surfaces along different grid directions
→ direction dependent branching behavior

* ... to be honest →



growth in **any** off-grid direction
only for short distances
(but better in 3D)

→ **conclusion on CA:**

fast (1% of CPU of Phase Field)
grid anisotropy reduced,
but essentially unavoidable
secondary arms mostly unrealistic

conclusion on phase field:

large community
far developed
precise
slow
low (but non-zero) grid anisotropy

→ new attempt: meshless method



”a computer algorithm is most benign on regular or almost regular grids“
(experienced modeller)

commonly used method:

start with regular grid

displace each node by small amount

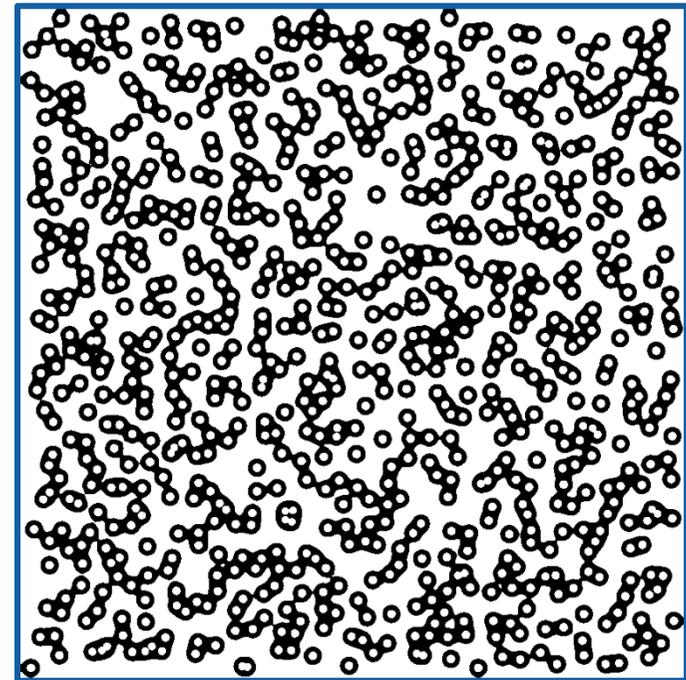
→ retains neighborhoods from cartesian grid

→ retains bookkeeping

known problems:

numerical instabilities

local divergence



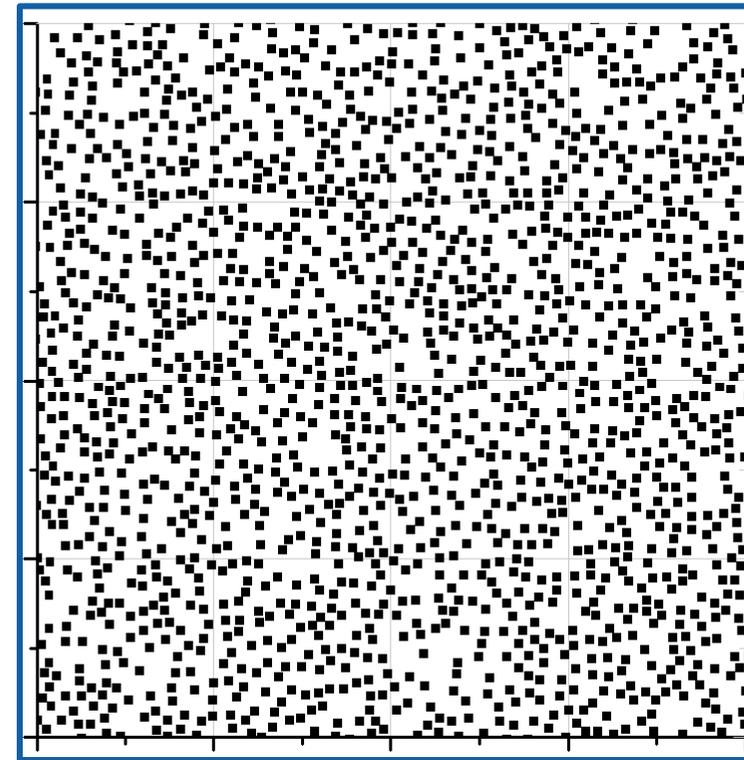
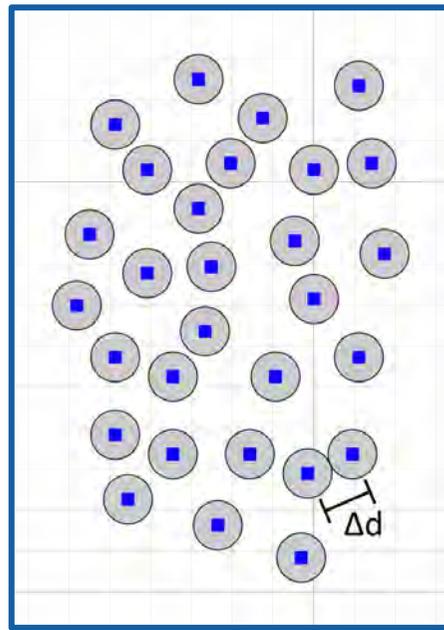
(Perko, CMES 2007)



random positioning of nodes

minimum distance Δd between every node for improved homogeneity

→ no preferred direction, isotropic at length scales $>\Delta d$



(Reuther, Sarler, Rettenmayr 2012, IJTS)

point based solver of partial differential equations

"Point Automata"

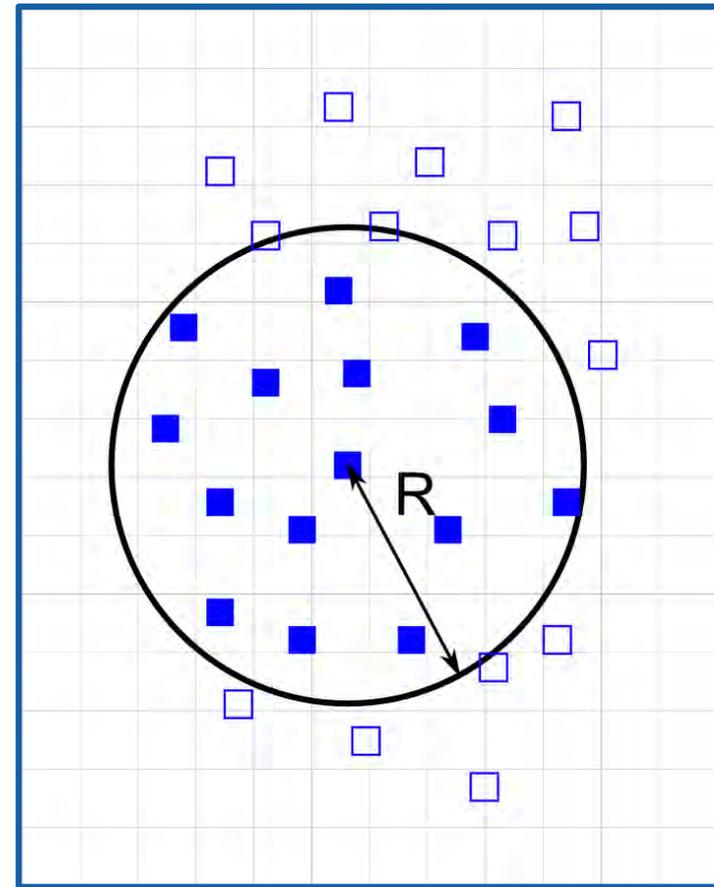
example: diffusion equation

interpolation of the concentration field
by a **distance weighted least squares fit** within radius R

Taylor series of 2nd degree:

$$c(x, y) = a_0 + a_1x + a_2y + a_3x^2 + a_4y^2 + a_5xy$$

$$\nabla^2 c = 2(a_3 + a_4)$$





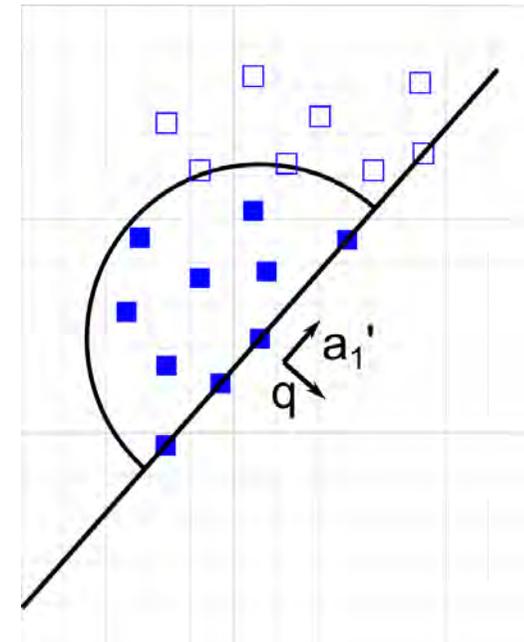
e.g. Neumann boundary conditions

$$c(x, y) = a_0 + a_1x + a_2y + a_3x^2 + a_4y^2 + a_5xy$$

$$\frac{\partial c}{\partial n} = n_x(a_1 + ya_5) + n_y(a_2 + xa_5) \equiv q$$

find c such that the coefficients from the least squares fit satisfy the boundary condition

tangential flow not treated explicitly at boundary



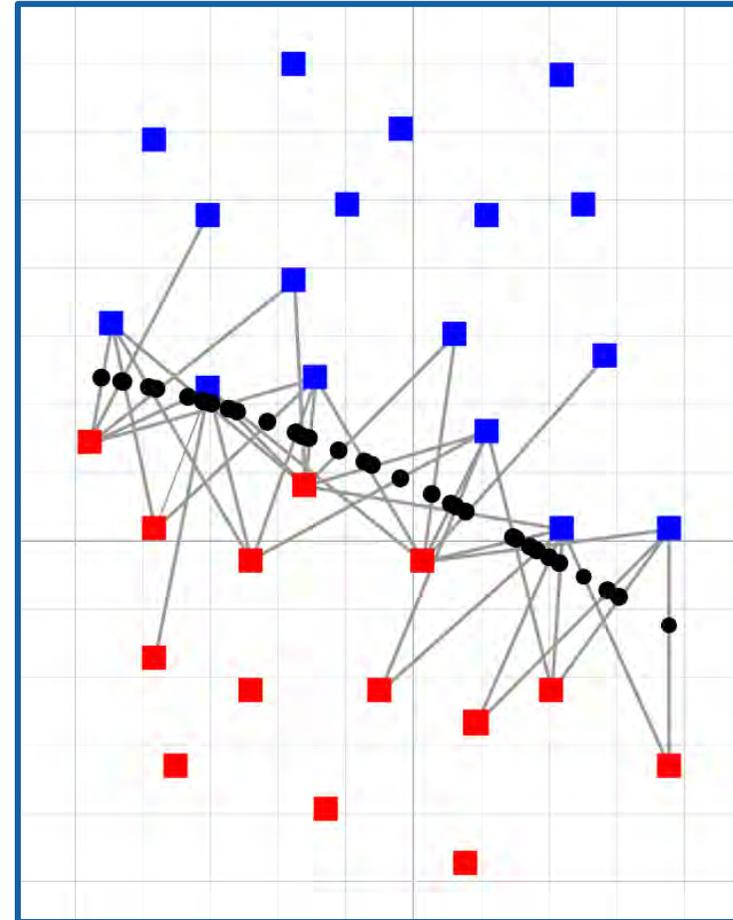
$$c(x, y) = a_0 + a_1'(n_yx - n_xy) + q(n_xx + n_yy) + a_3x^2 + a_4y^2 + a_5xy$$

normal term: given from boundary condition

tangential term: included in the fit



- grid node ■:
solid or liquid
- interface ● :
"particles" between nodes
with different state

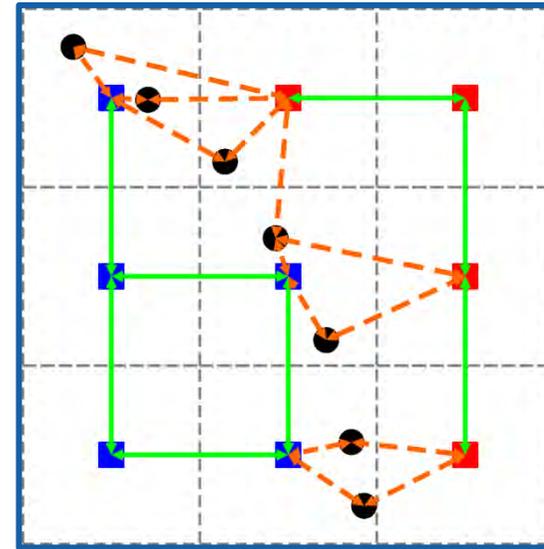


Reuther, Rettenmayr, Acta Mater 2013



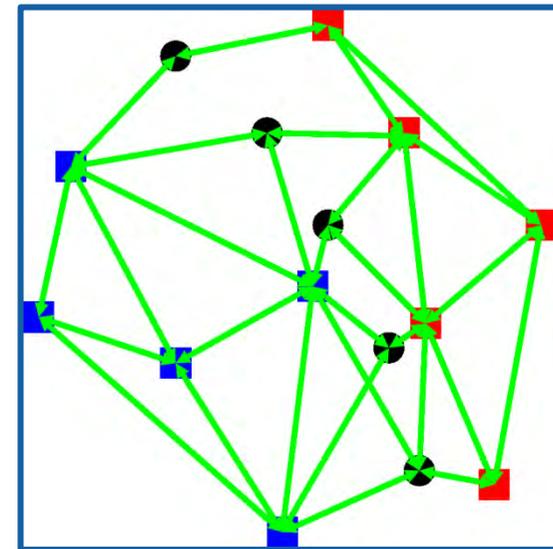
”particles“ on a fixed, regular grid:

→ interpolation/extrapolation scheme
outside finite difference method



particles on an irregular grid

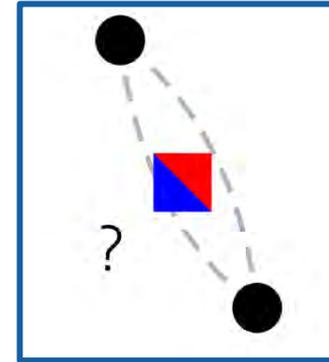
→ interpolation scheme is inherent
to point based meshless method





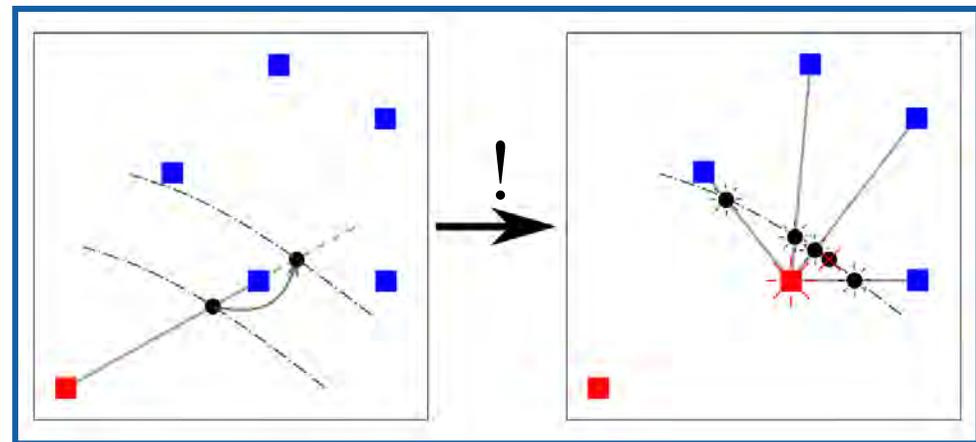
particle movement in point automata:

introduction of "free" particles requires special attention for the node bookkeeping

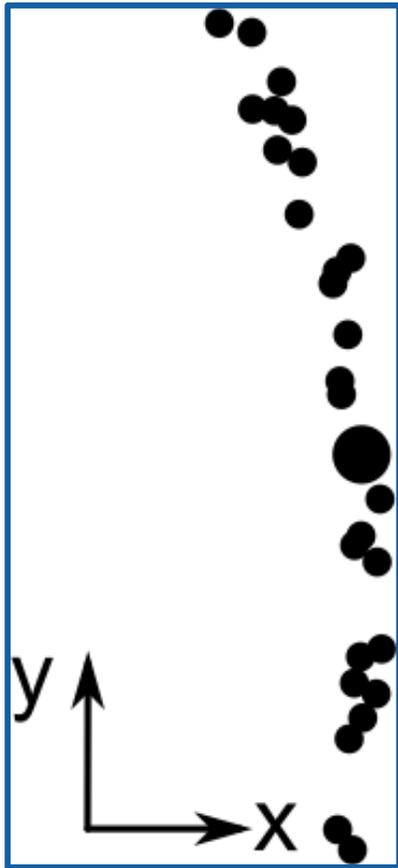


→ particle-node coupling trapping:

new particles introduced
old particle deleted



total least square fit



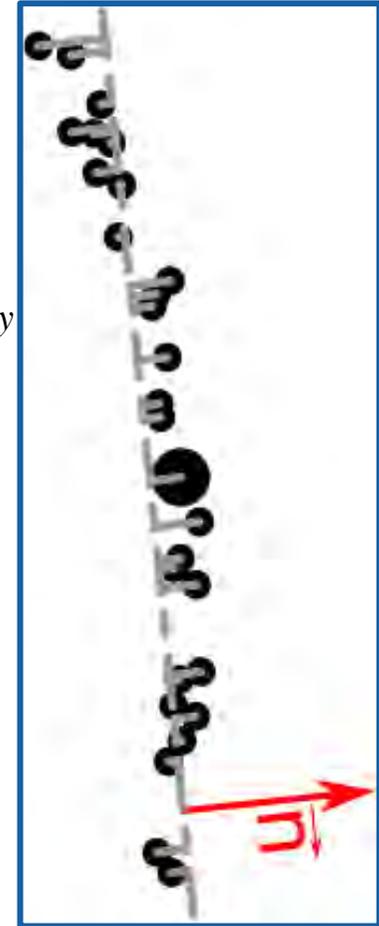
$$n = \begin{pmatrix} 1 \\ a \end{pmatrix}$$

$$a = \frac{1}{2} \Sigma_{yy} - \Sigma_{xx} \pm \sqrt{(\Sigma_{yy} - \Sigma_{xx})^2 + 4\Sigma_{xy}^2} / \Sigma_{xy}$$

with

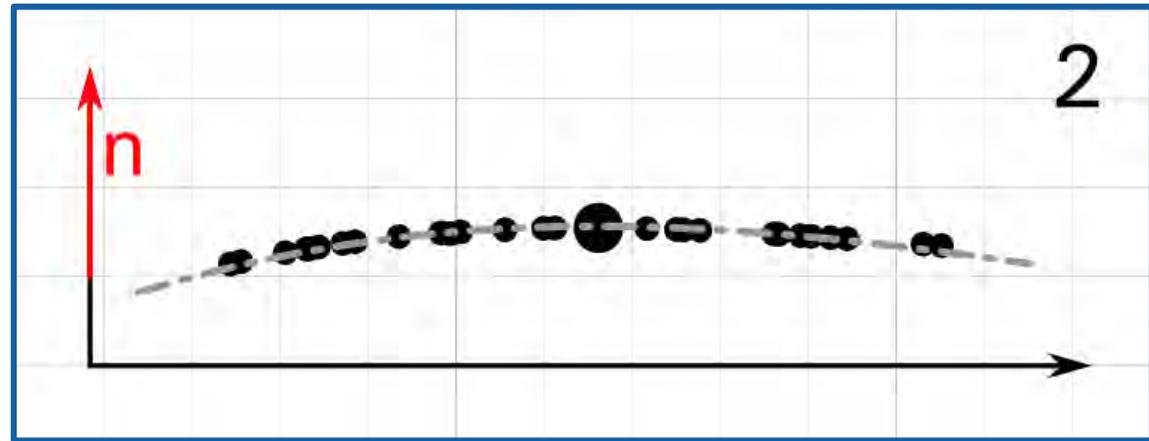
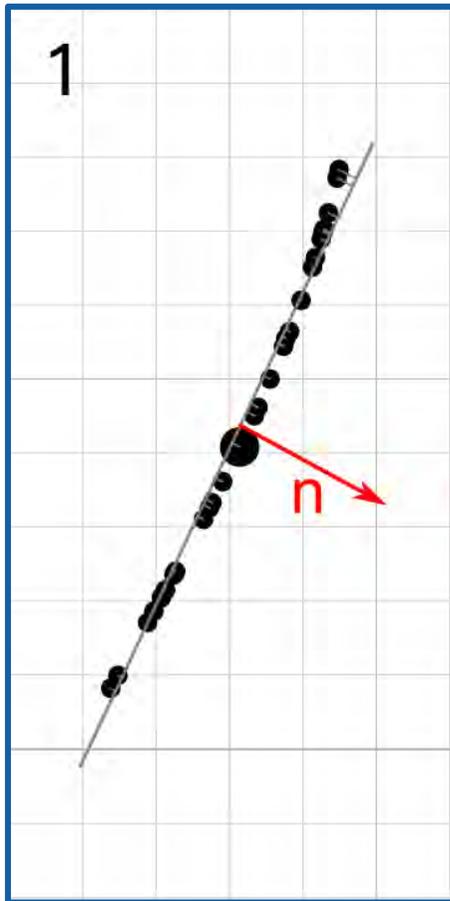
$$\Sigma_{xy} = \sum_i w_i x_i y_i$$

etc.



least square fit

in the cartesian coordinate system



$$\Delta T = \Gamma(\theta) \cdot K$$

local undercooling dependent on

interface geometry:

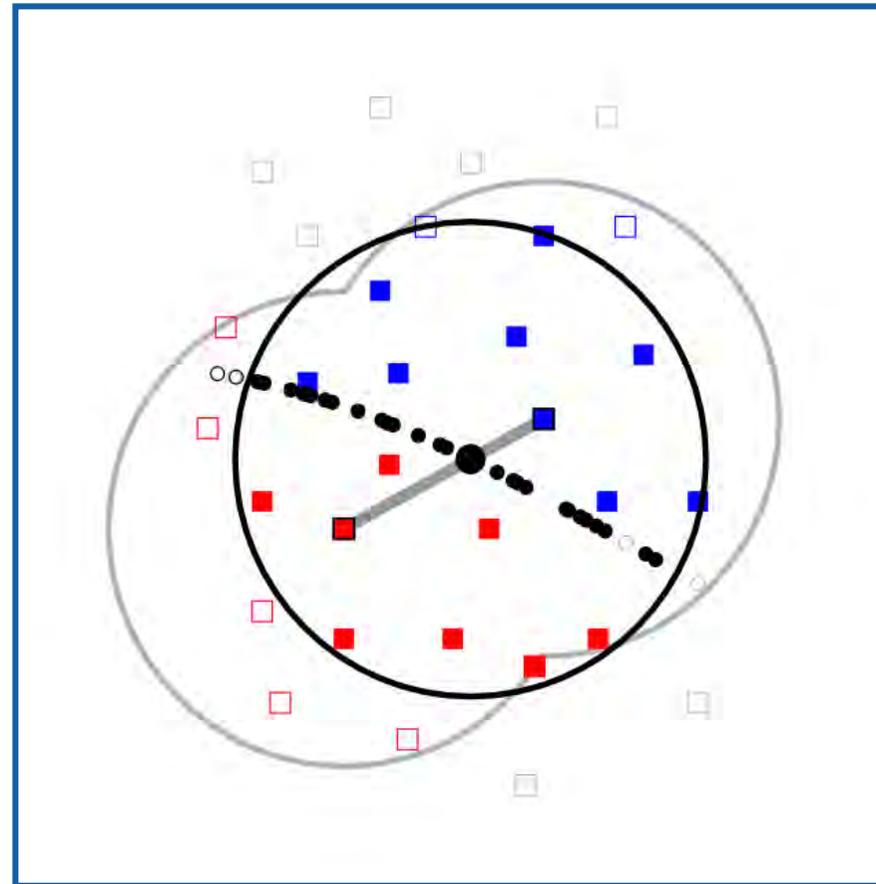
curvature (K)

angle (θ) to normal direction

interface **velocity**

defined by mass balance

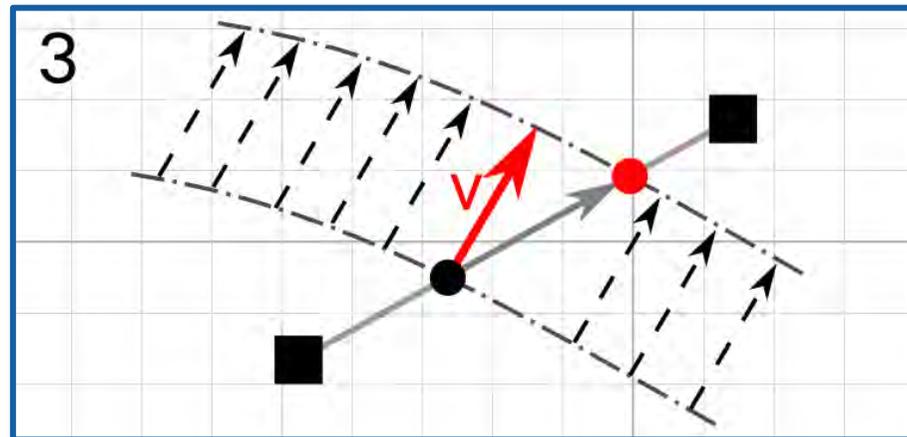
$$\vec{v} \cdot (c_l^* - c_s^*) = D_l \frac{\partial c_l}{\partial \vec{n}_{l \rightarrow s}} + D_s \frac{\partial c_s}{\partial \vec{n}_{l \rightarrow s}}$$



Reuther and Rettenmayr, Acta Mater. 2013



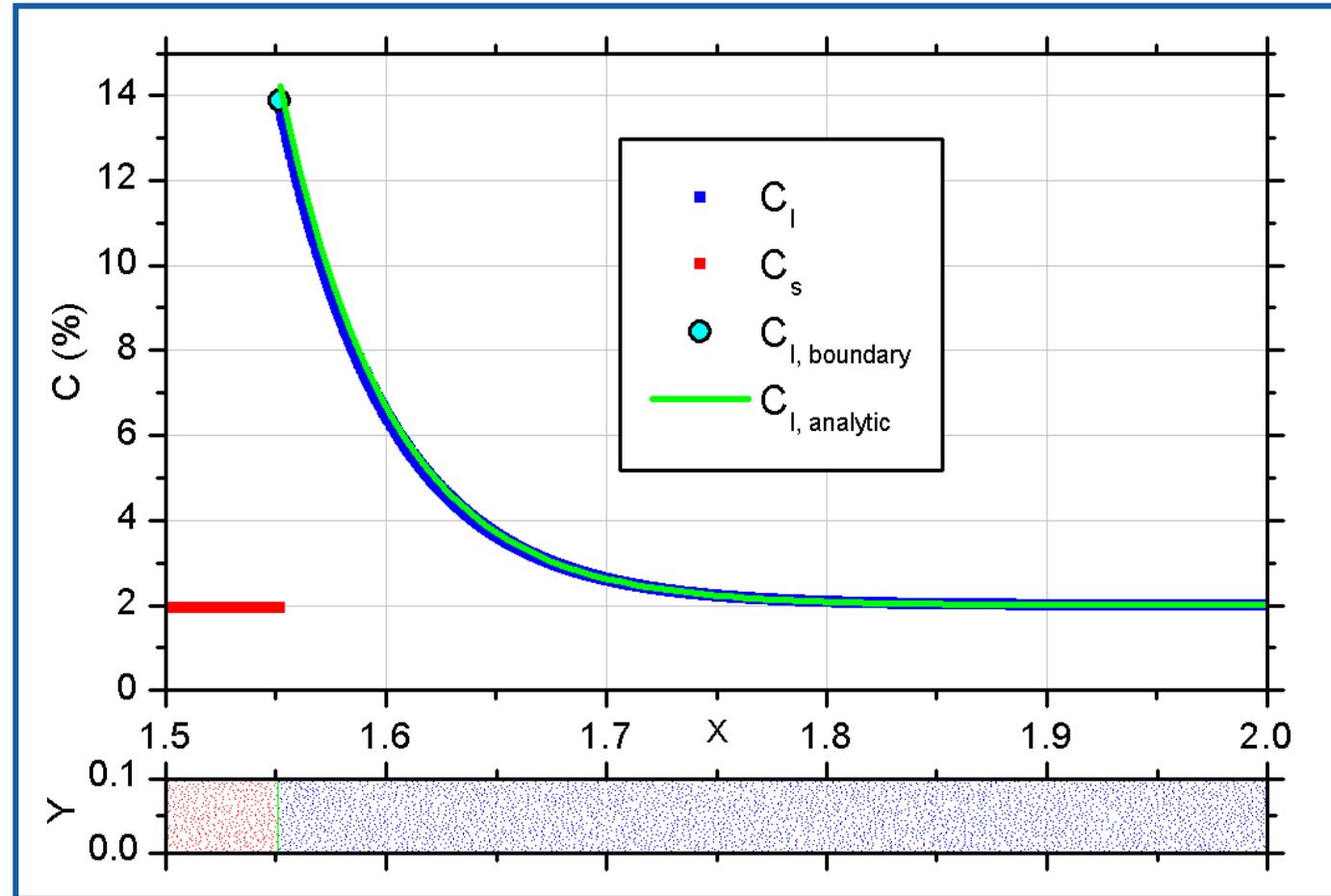
interface migration **direction**
defined by translation of the fit





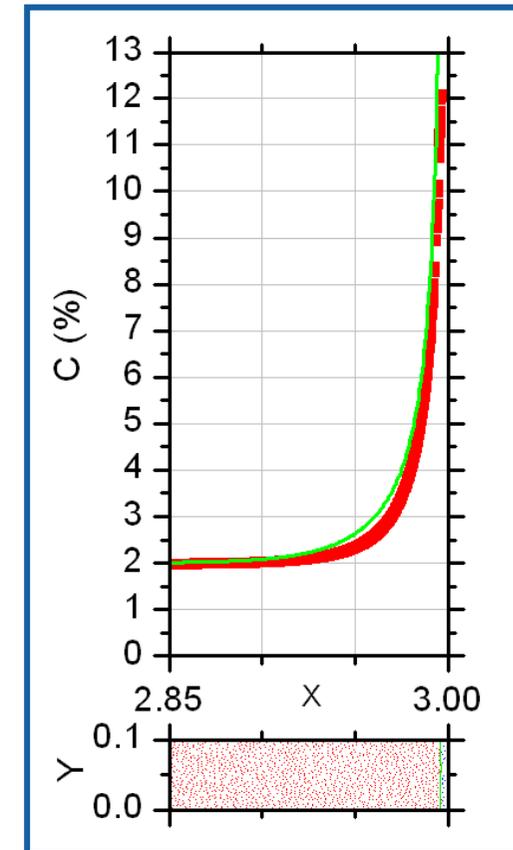
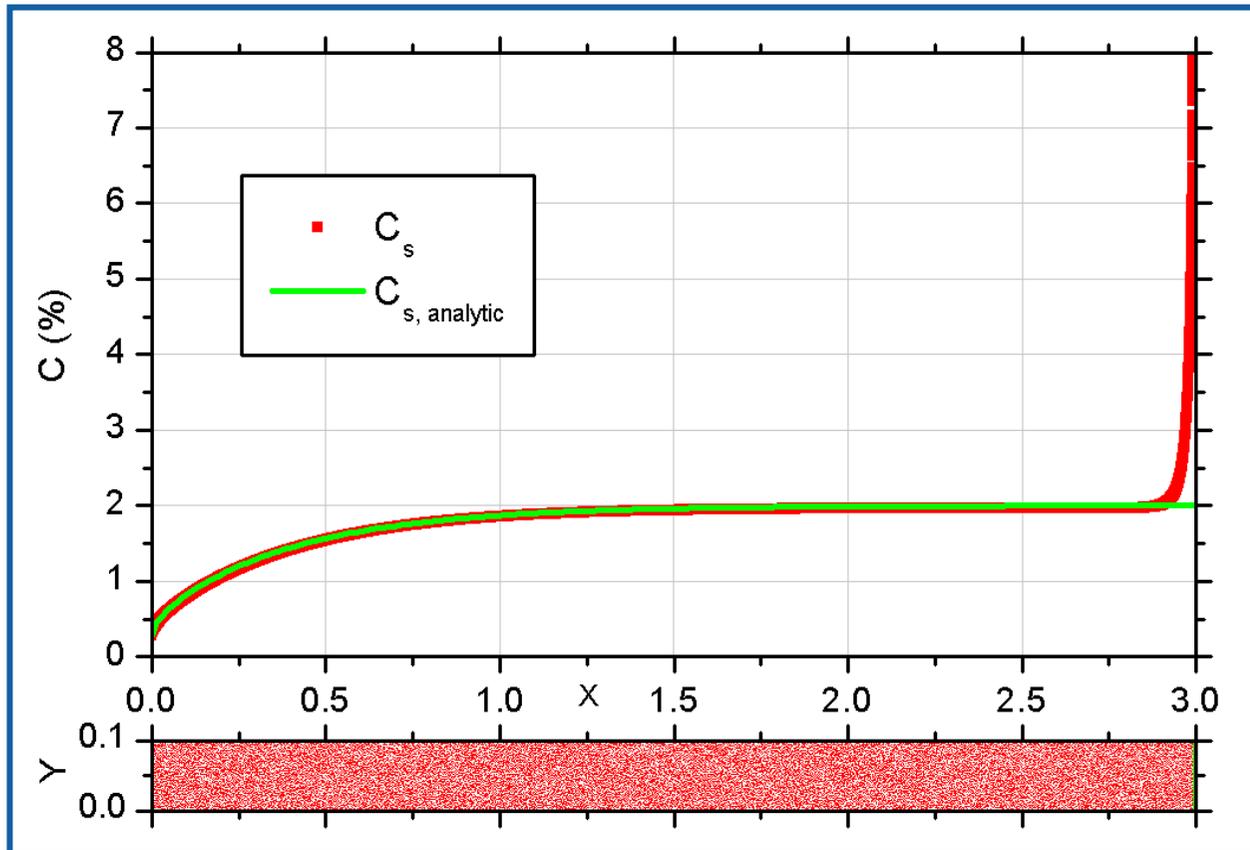
steady state solidification, concentration profile in the liquid

$$C_0 = 2\%$$
$$k = 0,14$$



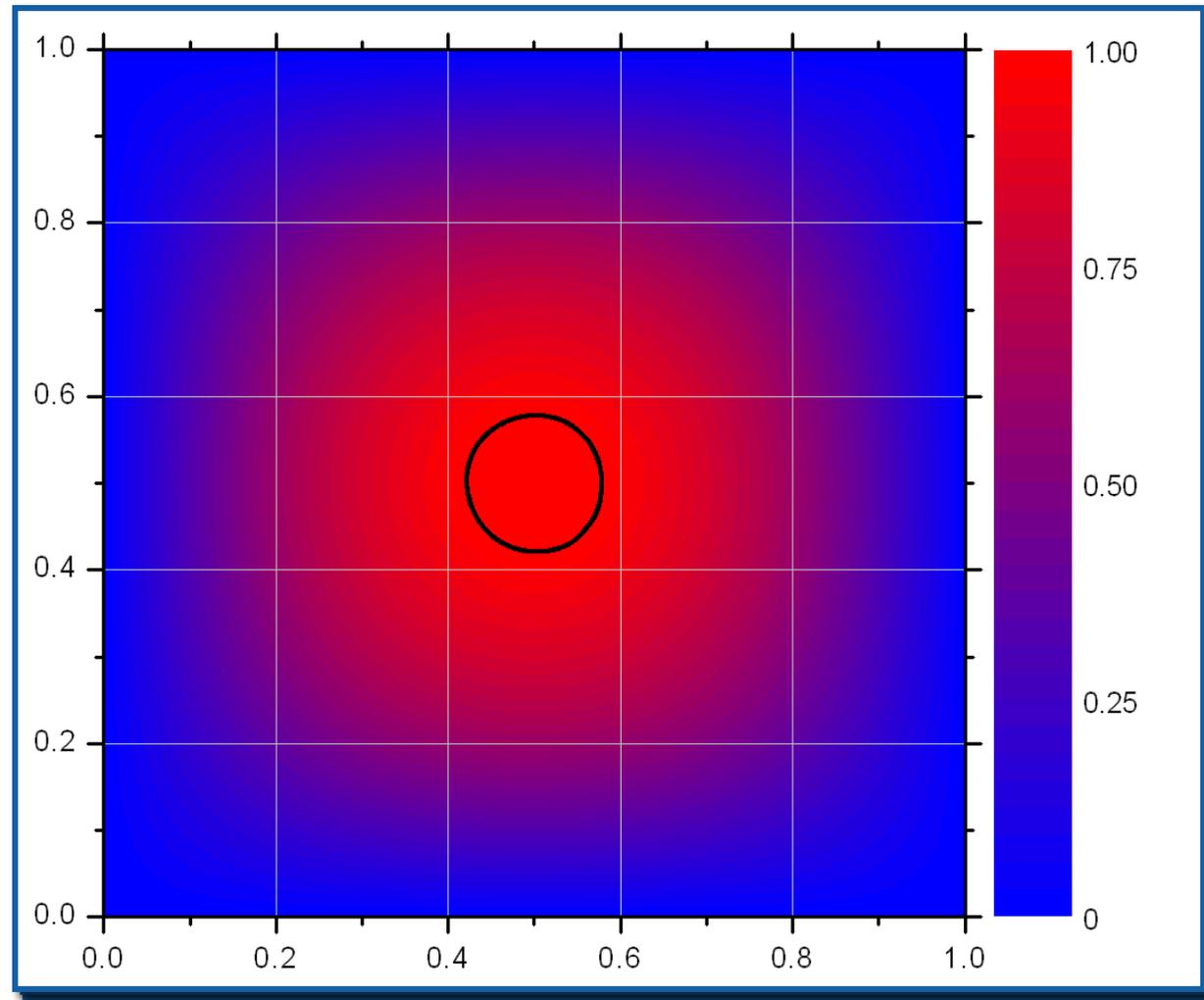


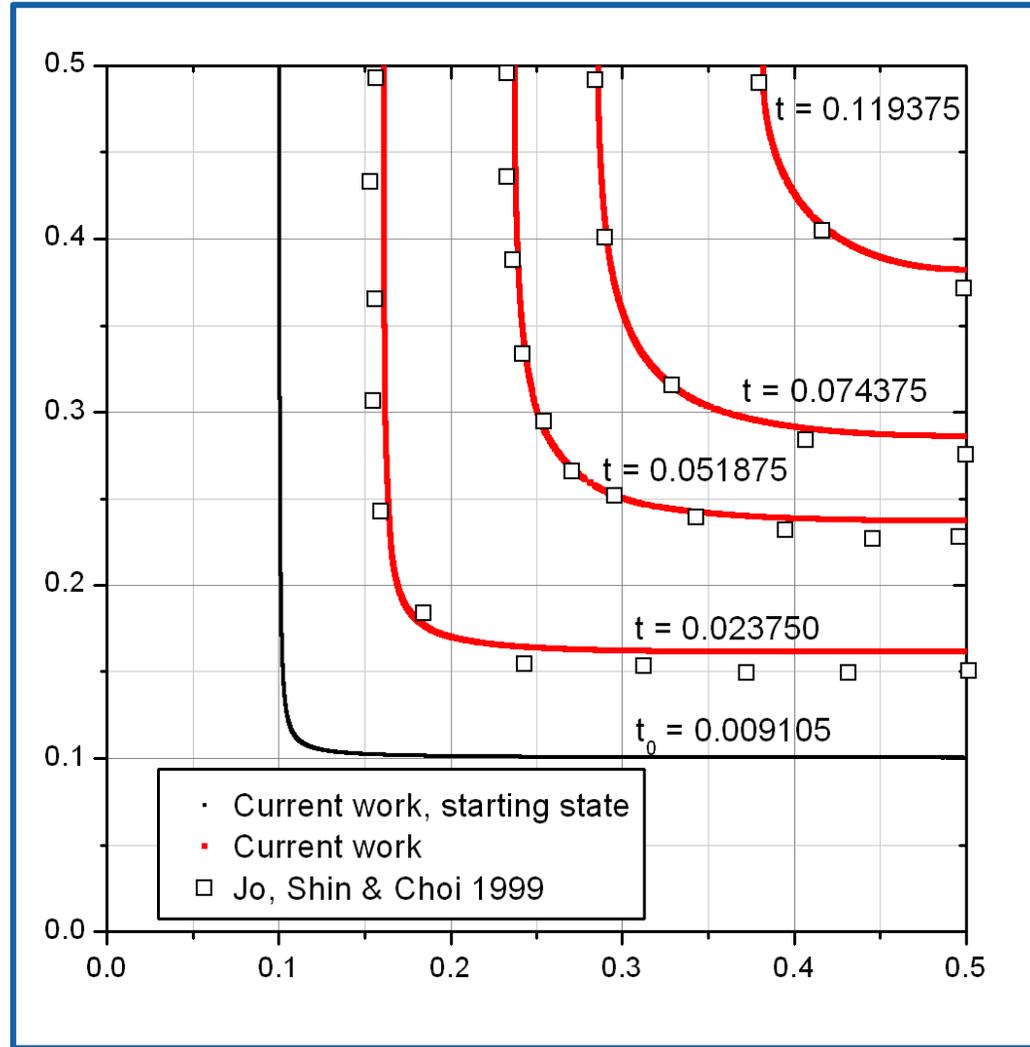
steady state solidification, initial and final transients

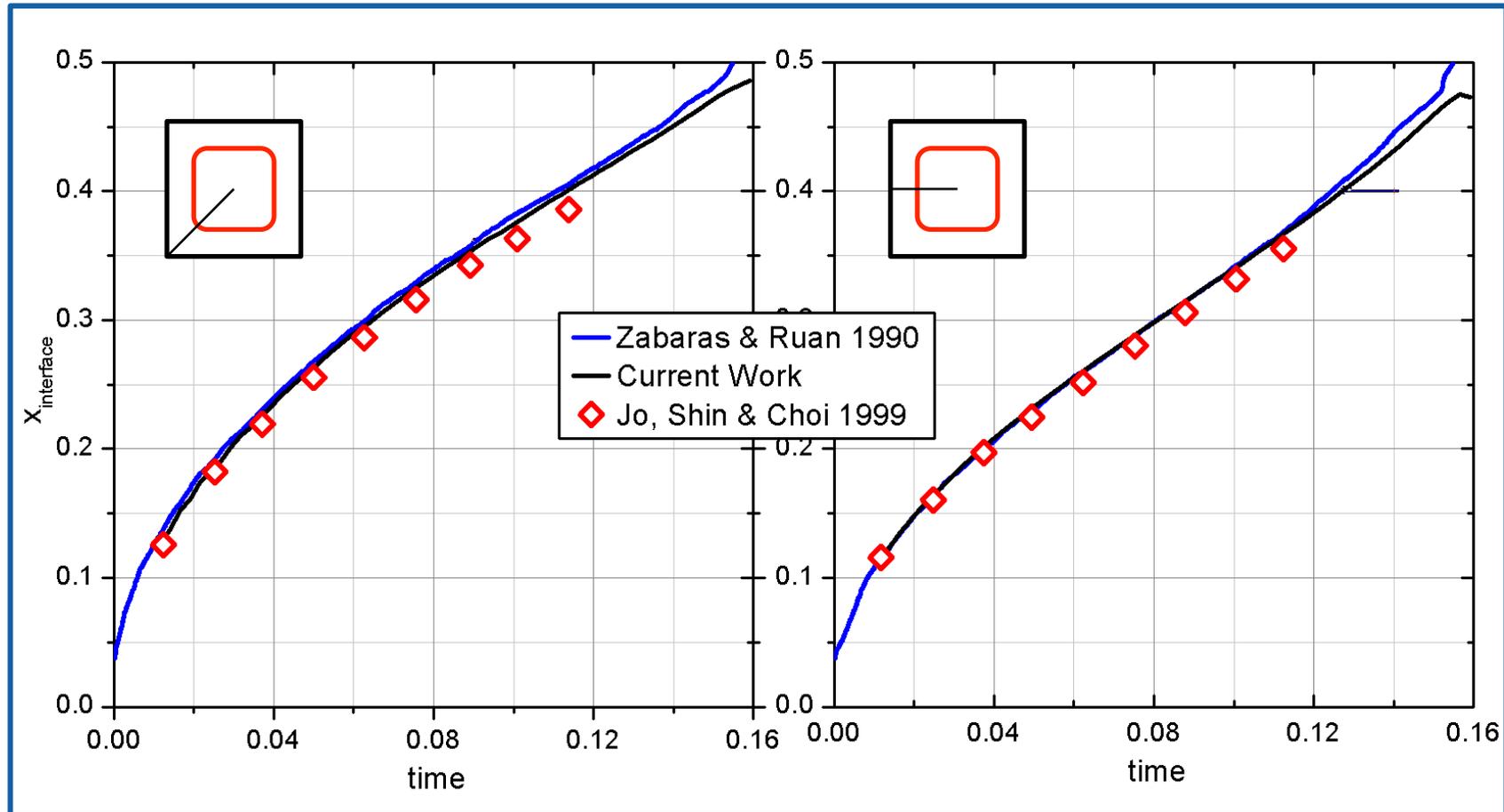


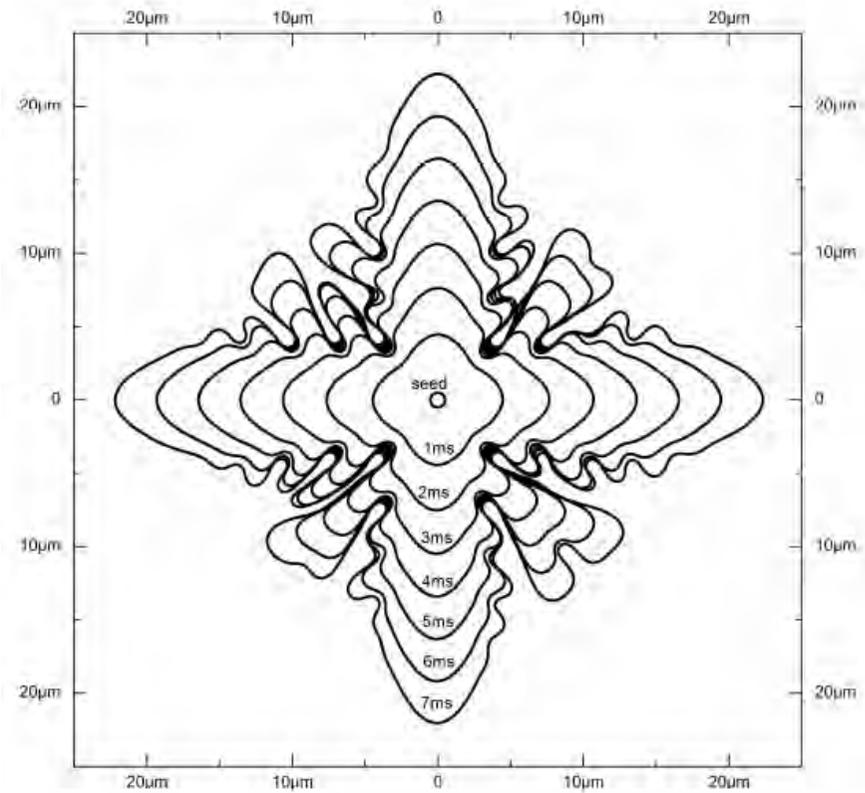
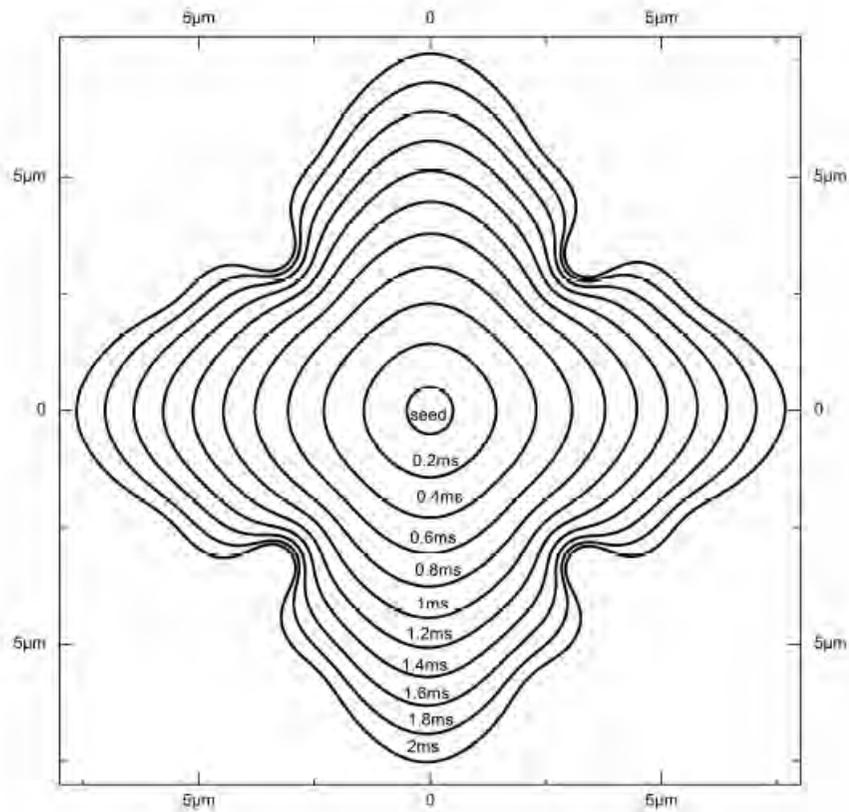


inward solidification in a square mold











simulation of complex morphologies with Cellular Automata

- fast (\rightarrow 100times faster than Phase Field)
- similarities with real growth morphologies undeniable

simulation of complex morphologies with Point Automata

- not very slow (\rightarrow 10 times faster than Phase Field)
- excellent reproduction of growth morphologies

simulation of complex morphologies with Phase Field

- far developed
 - slow (\rightarrow not faster than Phase Field)
 - very good reproduction of growth morphologies
-